

Preparation of PM Fe-FeAl and Fe-Fe₂Al₅ Composites

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Abstract

Intermetallic iron aluminides are promising for high temperature engineering applications due to their excellent elevated temperature strengths and high corrosion resistance. Iron aluminides prepared by pressureless sintering of blended Fe and Al powders, however, cannot be used for structural applications directly as the materials contain crack and porosity. In this paper, the pressureless sintered FeAl and Fe₂Al₅ powders were used as reinforcements in Fe-base composites. They were mixed with pure Fe powders with varied iron aluminide contents of 10, 20 and 30 wt.%. The compacts of mixed iron aluminide and Fe powders were sintered under controlled atmosphere at 1150 °C for 45 minutes. Compared to non-reinforced sintered Fe specimens, the sintered Fe composites with 10 wt.% iron aluminide reinforcement showed improved tensile strengths and hardness. Density of the aluminide-reinforced Fe composites were lower than that of the non-reinforced sintered Fe specimens. This suggests that the aluminide-reinforced Fe composites have better strength to weight ratio.

Keywords: Iron aluminides, sintering, composites

1. Introduction

Powder metallurgy (PM) is considered as a promising processing method for producing particulate-reinforced metal matrix composites (MMCs) as it provides benefits including microstructural homogeneity, materials saving and near net shape characters. Our attempt to

produce this type of materials started with the 316L-ceramics system [1]. In our first work, three ceramics including Al₂O₃, SiC and TiC powders with particle sizes of less than 20 micron were used as reinforcements for the 316L matrix. The mixtures of ceramics and 316L powders were compacted and sintered using a conventional P/M

process. The sintered Al_2O_3 -reinforced 316L material exhibited inferior sintered density, strengths and ductility but superior hardness to those of the sintered 316L material. Strength decrease in the sintered Al_2O_3 -reinforced 316L material was attributed to bad wettability of 316L matrix on Al_2O_3 particles. Addition of SiC particles to 316L powders seemed to improve sintered density and mechanical properties. This was attributed to wetting of boundaries between 316L grains and reinforcement SiC particles caused by melting materials during sintering process. Addition of up to 1.5 wt. % TiC improved the sintered TiC-reinforced 316L material properties. In contrast, addition of too high TiC particle contents resulted in that some coarse TiC particles occupied in boundaries between 316L grains. This caused detrimental effect on mechanical properties of the sintered TiC-reinforced 316L material. Experimental works had been carried out to understand deeply on particulate-reinforced MMCs by using Fe as a matrix, Al_2O_3 as an inert ceramic and SiC as a reactive ceramic reinforcement [2, 3]. The reactive SiC particle decomposed and reacted with Fe at temperatures higher than 1200 °C. Because of the reaction, different kinds of carbide-reinforced Fe composites were studied [4-7]. It was found that stability of carbide contacting with Fe at elevated temperatures was the key factor controlling reaction between carbide particles and Fe matrix.

All the reinforcements investigated as given above failed to provide integrity with Fe matrix. In this work, new reinforcements of iron aluminides were chosen to prepare sintered Fe base composites, not only because of iron

aluminide potential set of physical, chemical and mechanical properties [8, 9] but also their low cost, low density, ease of fabrication, good wear resistance and resistance to oxidation and corrosion [10, 11]. The aluminides (FeAl and Fe_2Al_5) were synthesized using pressureless sintering [12]. The as-synthesized aluminides were ground and mixed with Fe powders and processed via a 'press and sinter' method.

2. Experimental procedure

Syntheses of Fe-Al and Fe_2Al_5 intermetallics were given in [12]. The as-synthesized materials were ground into powdered forms with particle sizes < 75 μm . The FeAl and Fe_2Al_5 powders were mixed with pure Fe powders with varied iron aluminide contents of 10, 20 and 30 wt.% as shown in Table 1. The compacts of mixed powders were sintered under control atmosphere at 1150 °C for 45 minutes. Microstructures and mechanical properties of the aluminide-reinforced composites were observed and tested, respectively.

Table 1 Nominal composition of the iron aluminide-reinforced Fe composites

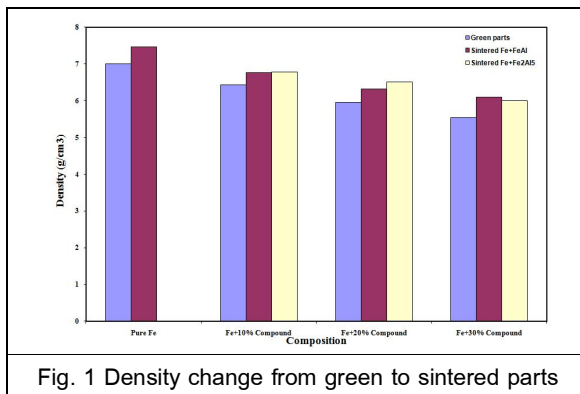
Composite	Iron aluminde	
	FeAl	Fe_2Al_5
1	Fe+ 10% FeAl	Fe+ 10% Fe_2Al_5
2	Fe+ 20% FeAl	Fe+ 20% Fe_2Al_5
3	Fe+ 30% FeAl	Fe+ 30% Fe_2Al_5

3. Results and Discussion

3.1 Density change

All sintered composites showed density increase from green to sintered specimens. This

indicates that all powder compacts gain densification by sintering. Addition of higher iron aluminide contents resulted in lower sintered materials density (Fig. 1). This is attributed to low density of iron aluminides. Decrease of sintered density due to iron aluminide addition is similar to the case of Al_2O_3 addition to Fe base composite [2, 3].



3.2 Mechanical properties

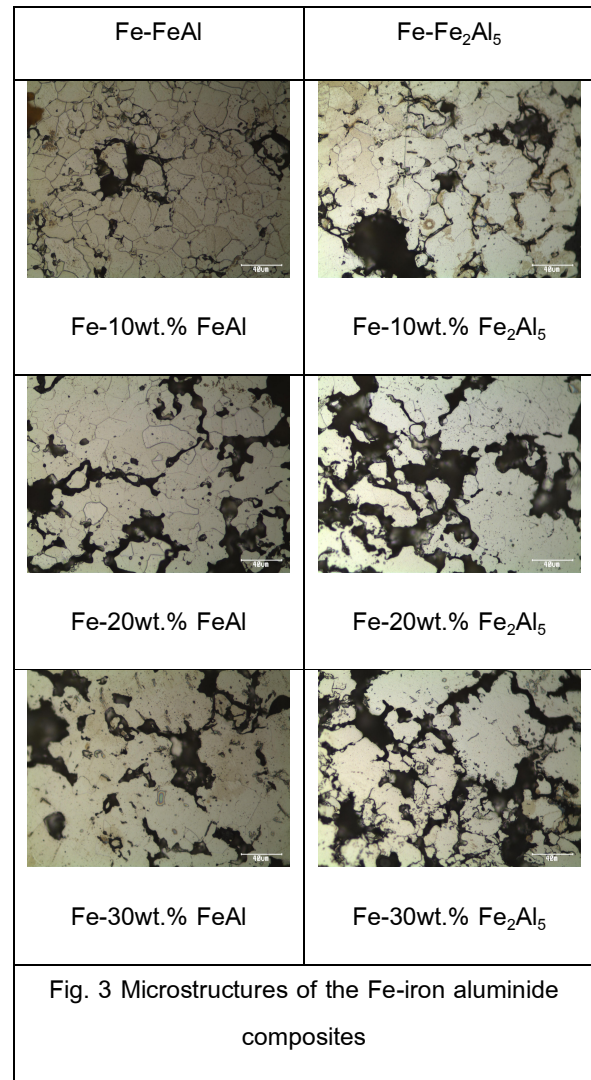
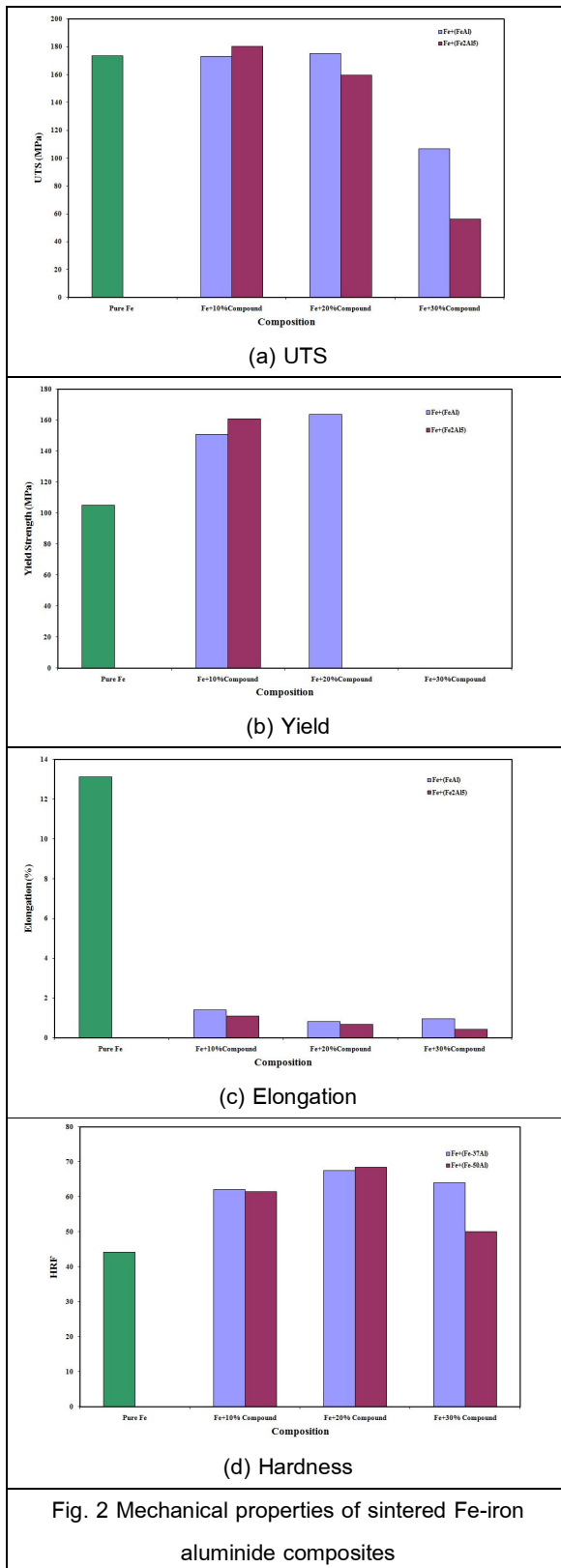
Only the 10 wt. % iron aluminide-reinforced Fe composite showed tensile properties (Fig. 2). Compared to the sintered pure Fe material, the iron aluminide-reinforced Fe composite exhibited higher tensile strengths but lower ductility although its microstructures showed that the reinforcement distribution was not uniform (Fig. 3). Increase of tensile strengths in the iron aluminide-reinforced Fe composite indicates benefit of aluminide addition to Fe matrix. Compared to the Al_2O_3 -reinforced Fe composite [2], the iron-aluminide-reinforced Fe composite showed positive improvement of tensile strengths. Experimental results indicated that the iron aluminide content added for preparation of the Fe base composite was limited at a certain value. Poor tensile properties of the iron aluminide-

reinforced composites with 20 and 30 wt. % iron aluminide additions were evidences of the limitation. Inferior mechanical properties of the iron aluminide-reinforced composites with too high contents of reinforcement were clearly related to the composite materials microstructures (Fig 3). Distribution of aluminide powder particles was not uniform in the cases of the composites with too high aluminide contents. Agglomerated iron aluminide particles occupying the boundaries between sintered Fe particles were clearly seen in the microstructures. The iron aluminide agglomeration along the boundaries inhibited sintering between the Fe particles to form Fe matrix.

Microstructural inhomogeneity may be attributed to improper powder mixing/blending and coarse aluminide powder particle size. Improvement of powder mixing and aluminide grain reduction will be carried out in the further work.

4. Conclusions

When compared to the non-reinforced sintered Fe specimens, the iron aluminide-reinforced Fe composite with 10 wt. % iron aluminide reinforcement showed improved tensile strengths and hardness although its microstructures showed that the reinforcement distribution was not uniform. Density of the composite was lower than the non-reinforced sintered Fe specimens. This suggests that the composite has better strength to weight ratio. However, this composite needs to be improved as it shows low ductility.



5. Acknowledgement

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6. References

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